Compatibility

Study of the Compatibility in the Poly (Vinyl-Chloride) -Poly (Vinyl-Alcohol) System

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Summary

Alfhough poly (vinyl-chloride) - PVC - and poly (vinyl-alcohol) - PVA mixtures are used in practice for coatings, containers, fibers, few studies are encounterd concerning their compatibility $(7, 8)$.

This paper establishes both the optimum compatibility ratio and the thermal behaviour of the mixtures.

The compatibility study in solid state was realised using the methodes: optical microscopy, thermogravimetry, DSC, inverse gaschromatography and IR spectroscopy.

EXPERIMENTAL

Materials, PVC was obtained by suspension polymerization. The molecular weight is $M_w = 57,000$.

PVC was purified by extraction with petroleum ether, for eight hours in a Soxhlet apparatus.

PVA was obtained by the basic hydrolysis of poly (vinyl acetat). Two PVA samples were used. The PVA₁ - (the first sample) has a polymerization degree of 1600, the PVA₂ - (the second sample) of 500. Both samples have the same hydrolysis degree 98%.

Mixture films, with mixture ratio of $10/90$; $30/70$; $50/50$; $70/30$; $80/20$ were obtained by polymer solubilisation in dimethyl sulfoxide (DMSO) common solvent.

A lo% polymer mixture solution was deposed in thin layers on glass lamella. The solvent was evaporated at $60 - 70$ °C. The so obtained films were extracted with methanol to eliminate last solvent traces.

Investigation methodes, The compatibility study in solid state was realised with the following methodes: optical microscopy, thermogravimetry, DSC, inverse gaschromatography and IR spectroscopy.

The microscopic examination of the films was executed using an IOR MC-1 microscop, in phase contrast, magnification 400 .

Recording conditions for TG curves were: MOM Budapest derivatograph,

heating rate 10° C/min., polymer sample weight for decomposition was of $2o$ mg.

The polymer mixtures were obtained by precipitation from common solution with a mixture of 40% methanol, 40% acethone and 20% ethanol. From TG curves the temperatures for different weight losses were determinated.

Activation energy of the first thermal decomposition (200-370 $^{\circ}$ C) step. was calculated using the Freeman - Carroll (3) method.

The experimental errors were appreciated with the least squares method. DSC curves were recorded with a Perkin - Elmer DSC2 using a heating rate of 32°C / min and polymer weight of 20 mg. From these curves, were evaluated the glass transition temperatures (T_{g}) and those of the decomposition.

T values were compared with those obtained by inverse gaschromatography, according to Guillet's method. (5) ,

Chromatographic columns were prepared in the following way: pure polymers and polymer mixtures in DMSO solution were deposed on the W Chromosorb by continuously agitation.

After solvent evaporation and vacuum drying, a chromatographic column with $1, 52$ m in lenght and 4 mm in the interior diameter was filled. The chromatographic columns were activated by heating in an inert gas (Argon) stream.

Toluen was used as "molecular probe".

The variation of the toluen retention time, $\log t_{\rm R}$, versus reciprocal temperature was recorded.

The glass transition temperature, $T_{g^{\phi}}$ was determined by the inflexion point of these curves.

For pure polymers T_{\perp} data were compared with literature values. The concordance was satisfactory: $T_{g, DV} = 78^{\circ}C_{\bullet} T_{g, DVA} = 60_{\bullet} 4^{\circ}C$ (1)

IR spectra were recorded with the Perkin-Elmer spectrometer 625 on films of aproximative constant thickness of $o_{\bullet} o_{\bullet}^2$ mm. For optical density comparison of the 332o cm \hat{O} -H) and 61o cm \hat{C} -C1) bands, internal standards bands in polymer mixtures were used. These are 295o cm⁻¹ and 2895 cm⁻¹ bands and 630 cm⁻¹ and 69o cm⁻¹ bands respectively.

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Microscopic aspect of pure and mixture polymers films are presented in Fig. 1.

It may easily be noted that for mean mixture ratios the phase separation is clear. The extreme mixtures look like an advanced dispersion. The thermodynamic incompatibility is evident.

Consequently, the time coexistence in relatively homogenous mixtures should be conditioned only by the limit phase interactions.

80/20 70/30 50/50 30/70 20/80 10/90 **PVC** $90/10$ PVA-1

Fig. 1. Microscopic aspects of PVC. PVA, and their mixtures in different ratios, Magnification 4oo

The existence of these interactions is emphasized by the thermal characteristics modifications, as presented in Tab. 1.

It may be remarked that mixtures with 70% - 20% PVA and 30% - 80% PVC show two thermal decomposition processes in the same temperature range. The temperatures corresponding to the maximum reaction rate are different from those of the pure components or from those calculated by additivtty rule. The experimental weight losses are smaller in all cases. The diminution may be caused by some cross-linking reactions along with the dehydroclorinatlon of PVC or the dehydration and degradation of PVA.

Generally, the mixtures show multiple glass transition temperatures smaller than T_{σ} of pure components.

By inverse gaschromatography, which is more sensible than the other methods, a suplimentary glass transition temperature T_{σ} , observed at smaller temperature.

This may be caused by the chain segment movements in the interphasic range where the mobility is higher. (4)

The compatibility diagram including the two coordinates: activation energy $-$ for the first thermal decomposition step $-$ and composition $(Fig, 2)$, evidences the fact that the components present a pseudocompatibility caused by the functional groups interactions for $90\% - 60\%$ PVA_p and 10% - 40% PVC. The pseudocompatibility range for a PVA with a smaller molecular weight extends up to 40% PVA₂ - 60% PVC.

Thermal characteristics of PVC/PVA mixtures. Thermal characteristics of PVC/PVA mixtures.

CI - inverse gaschromatography CI - inverse gaschromatography

Fig. 2. Compatibility diagram for PVC/PVA system.

Functional groups interactions were evidence by IR spectroscopy for the mixture ratios mentioned above.

The optical densities have more different vaIues than those evaluated if the independent behaviour hypothesis of the components would be observed.

CONCLUSIONS

1. From a thermodynamic point of view PVC/PVA mixtures are incompatible.

2. In 90% - 50% PVA - 10% - 50% PVC composition range a pseudocompatibility caused by functional groups interactions appears. conferring the mixture some improved properties. Among these properties we mention: the increasing thermal stability along with the lowering of the glass transition temperatures.

This fact indicates a possibility for an easier processing.

3. At high temperatures, the components interact forming crosslinking structures.

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